# Growth, Structural and Micro hardness studies of KSbF<sub>4</sub> and K<sub>2</sub>SbF<sub>5</sub> crystals

C. Besky Job, R. Shabu, S. Paulraj

Abstract—Interest in Potassium Fluoro Antimonate crystals has been increased for the last four decades due its superionic conduction and its unusual electro-optic properties. Potassium tetra fluoro antimonate  $(KSbF_4)$  and Potassium penta fluoro antimonite  $(K_2SbF_5)$  crystals have been grown by slow evaporation method.  $KSbF_4$  crystallizes into orthorhombic structure with a space group Pmmn.  $K_2SbF_5$  belongs to orthorhombic crystal system with a space group Cmcm. Micro indentation analysis on these crystals indicates that they are moderately softer substances. Both crystals revealed reverse indentation size effect (RISE). Variation of stiffness constant with load has been discussed. Yield strength for  $KSbF_4$  and  $K_2SbF_5$  crystals have been found out as 16.72 and 16.941 MPa respectively.

Index Terms—Micro indentation, Potassium fluoro antimonate crystal, Stiffness constant, Yield strength.

#### I. INTRODUCTION

Although hardness has been defined in several ways it is now generally accepted that it is the resistance offered by the crystal for the movement of dislocations and practically it is the resistance offered by a material to localized plastic deformation [1]. The microhardness is a mechanical parameter that is strongly related to the structure and composition of the crystalline solids [2]. The hardness is estimated from the ratio of the load applied on indenter to the area of the impression left on the specimen. During the indentation process, the external work applied by the indenter is converted into the strain energy component proportional to the volume of the resultant impression and the surface energy component [3]. Hardness of the material depends on different parameters such as lattice energy, Debey temperature, heat of formation and interatomic spacing [4], [5]. It is well known that micro hardness of solids depends on the applied indentation test load. This phenomenon is known as the indentation size effect (ISE) [6]. In normal ISE we have been observed higher hardness values at lower loads and it decreases as the load is increased. It has been observed in a variety of materials like alkali halides [7] and alums [8]. In some cases, a different trend with initial increase of hardness

**C. Besky Job**, Department of Physics and Research Centre, Scott Christian College (Autonomous), Nagercoil, India, +91 9487026024.

**R. Shabu**, Department of Physics and Research Centre, Scott Christian College (Autonomous), Nagercoil, India, +91 9940844860.

**S. Paulraj**, Department of Physics and Research Centre, St. Xavier College (Autonomous), Palayamkottai, India, 0462-2578428.

with increase in load then followed by a decrease and finally hardness becoming independent at higher loads, known as reverse indentation size effect (RISE) [9]. Similar observations have been found out in our present study. KSbF<sub>4</sub> and K<sub>2</sub>SbF<sub>5</sub> crystals have aroused great interest because of their several interesting electro-optical and super ionic properties [10]-[12]. Micro hardness studies have been made on various fluoro antimonate crystals such as Sodium fluoro antimonate crystals [13], [14] and Ammonium fluoro antimonate crystals [15], [16]. Researchers have devoted many pages for finding the characterization of Potassium fluoro antimonate crystals. However in the literature, information on the mechanical properties of Potassium fluoro antimonate crystals is meager. Therefore in this paper we report the growth, structural and micro hardness studies of KSbF<sub>4</sub> and K<sub>2</sub>SbF<sub>5</sub> crystals.

# II. MATERIALS AND METHODS

Antimony tri fluoride forms several complexes with Potassium fluoride. Among the complexes,  $KSbF_4$  and  $K_2SbF_5$  crystals have been synthesized by reacting Antimony tri fluoride with Potassium fluoride in the appropriate molar ratio.

$$KF + SbF_3 \rightarrow KSbF_4$$
 (1)

$$2KF + SbF_3 \rightarrow K_2SbF_5 \tag{2}$$

The reactants were stirred well using magnetic stirrer to ensure uniform temperature and concentration throughout the entire volume of the solution. The filtered solution was transferred into a crystallizer and allowed to evaporate slowly under room temperature. Small transparent rectangular shaped crystals were obtained in a period of about fifteen days. The grown KSbF<sub>4</sub> and K<sub>2</sub>SbF<sub>5</sub> crystals are shown in Fig. 1 and Fig.2.

The X-ray diffraction patterns (XRD) of the powdered samples were obtained by using Philips X'pert Pro X-ray automatic diffractometer in the range of  $10-70^\circ$  with CuK $\alpha$  radiation of wavelength 1.54056 Å. The single crystal X- ray diffraction analysis of crystal was carried out using ENRAF NONIUS CAD-4 Diffractometer with MoK $\alpha$  radiation ( $\lambda$ =0.7107Å). Hardness measurements have been carried out using REICHERT MD 4000 E Vicker's Ultra Micro hardness tester attached to a metallurgical optical microscope. The well polished crystal was mounted on the platform of the microhardness tester and loads of different magnitudes were applied over a fixed interval time of 15 Sec. The average diagonal lengths of indentation for various loads were measured.



Fig. 1. Photograph of the grown KSbF<sub>4</sub> crystal

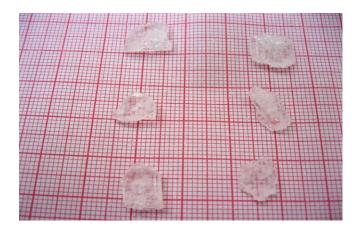


Fig. 2. Photograph of the grown K<sub>2</sub>SbF<sub>5</sub> crystal

# III. RESULT AND DISCUSSIONS

The observed indexed Powder XRD pattern for  $KSbF_4$  crystal is shown in Fig.3.

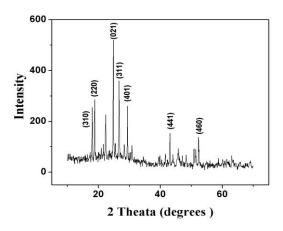


Fig. 3. Indexed observed XRD Pattern of KSbF<sub>4</sub> crystal

The XRD data match very well with the JCPDS file No. 85–0626, which confirms the identity of the grown crystal. The result of single crystal XRD indicates that KSbF<sub>4</sub> crystallizes into orthorhombic structure with space group Pmmn. The lattice parameters are determined to be a=16.28

Å, b=11.51 Å, c= 4.543 Å,  $\alpha$ = $\beta$ = $\gamma$ =90°, V=857.33 Å<sup>3</sup>. These results have been found to be in agreement with the already reported values [17]. Observed indexed Powder XRD pattern for K<sub>2</sub>SbF<sub>5</sub> crystal is shown in Fig. 4.

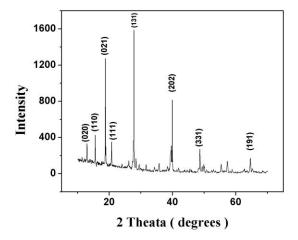


Fig. 4. Indexed observed XRD Pattern of K<sub>2</sub>SbF<sub>5</sub> crystal

The pattern is very well in agreement with the JCPDS file 77-0878, which confirms the presence of  $K_2SbF_5$  crystal structure. The result of single crystal XRD indicate that  $K_2SbF_5$  crystal belongs to orthorhombic crystal system, having lattice dimensions a=6.27Å, b=13.66 Å, c=6.49 Å and  $\alpha$ = $\beta$ = $\gamma$ =90°, V=556 ų with a space group Cmcm, which are in agreement with the already reported values [18]-[21].

The Vicker's micro hardness value (Hv) was calculated using the relation [6],

$$Hv=1.8544P/d^2 \text{ kg/mm}^2$$
 (3)

Where P is the applied load and d is the average diagonal length of the indentation impression. A plot of hardness value (Hv) with the applied load (P) is shown in Fig.5.

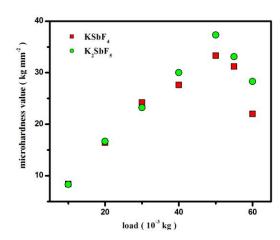


Fig. 5. Variation of microhardness with load for  $KSbF_4$  and  $K_2SbF_5$  crystals.

In the present study for both crystals the value of Hv increase as the load increase and finally decreases for higher loads (above 50gm), revealed reverse indentation size effect (RISE). Above 50 gm cracks were initiated on the crystal surface around the indenter. The major contribution to the increase in hardness is attributed to the high stress required

for homogeneous nucleation of dislocation in the small dislocation free region indented [22]-[24].

When the applied load is small, the indenter penetrates only the surface layers, consequently dislocations are nucleated along a particular slip plane near the surface and therefore the effect is shown sharply. When the penetration depth increases with applied load, nucleation of dislocation involves in another set of slip planes just below the indenter in the crystal. After a certain penetration, the effect of inner slip planes becomes more prominent than that of slip planes along the surface layers and hence the microhardness is independent on load [25].

It can also be explained in another way. During indentation the thickness of the distorted zone is limited for small loads, hence we observe a steady increase in hardness with load. As the depth of the indenter increases with load, the effect of the distortion zone decreases and hence the dependence of microhardness is less. For large loads (above 50 g), the indenter reaches the undistorted zone, hence the microhardness in independent on load [26], [27]. From the literature it has been found that the reverse ISE occurs only in materials in which plastic deformation is dominant [28], [29]. Investigations showed that all semiconductors exhibit RISE [30]. The same phenomenon is observed in the present insulating crystals.

The relation between the load and size of the indentation is given by Meyer's law [31],

$$P = K_1 d^n (4)$$

Where P is load, d is the diagonal length of impression,  $K_1$  and n are constants. Work hardening coefficient (n), a measure of the strength of the crystal, is computed from the slope of the log P versus log d plot (Fig.6) and  $K_1$ , the standard hardness is noted by the intercept of the graph.

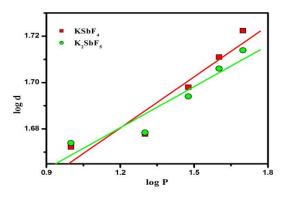


Fig. 6. log P versus log d Plot for KSbF<sub>4</sub> and K<sub>2</sub>SbF<sub>5</sub> crystals.

The value of n for  $KSbF_4$  and  $KSb_2F_5$  crystals have been found out as 1.59 and 1.63 respectively using least square fit method. Low work hardening coefficient shows less dislocation in the grown crystal, since work hardening coefficient is caused by the dislocations present in the crystal [32], [33]. From many observations it has been found that the value of n lies between 1 and 1.6 for hard materials and it is more than 1.6 for soft materials [34]. The values observed in the present study are just around 1.6 suggesting that  $KSbF_4$  and  $K_2SbF_5$  crystals are moderately softer substances. The value of n for  $K_2SbF_5$  is in close agreement with that of mixed crystals of  $(NH_4)_2SbF_5 - K_2SbF_5$  (n = 1.66) [35] and for

 $Na_2SbF_5$  (n = 1.5) [13]. The n value of  $KSbF_4$  is slightly differ with  $NaSbF_4$  (n = 1.3) [13] and suggest that  $KSbF_4$  is softer crystal than  $NaSbF_4$ . The decreasing n value of  $KSbF_4$  compared with  $NaBF_4$  (n = 1.79) and  $NH_4BF_4$  (n = 1.85) [36] may be attributed by the introduction of antimony in the place of Boron. From the n values, it has been found that  $K_2SbF_5$  crystal is softer than  $KSbF_4$  crystal.

Since the material takes some time to revert to elastic mode after every indentation, an additional correction factor x known as the measure of dislocation density of the material is applied to the observed d value [32] (Justin Raj et al 2008). Meyer relation may be satisfied as given below.

$$P = K_2 (d + x)^2$$
 (5)

Substituting for P from Meyer relation we get.

$$K_1 d^n = K_2 (d + x)^2$$
 (6)

$$d^{n/2} = (K_2/K_1)^{1/2} d + (K_2/K_1)^{1/2} x$$
 (7)

The correction factor x was determined from intercept of straight line obtained by plotting d versus  $d^{n/2}$  (Fig.7). Slope yields  $(K_2/K_1)^{1/2}$ , where  $K_2$  is a constant.

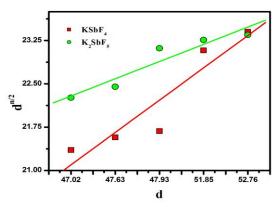


Fig. 7. Plot of d versus d<sup>n/2</sup> for KSbF<sub>4</sub> and K<sub>2</sub>SbF<sub>5</sub> crystals

The micro hardness value correlates with other mechanical properties such as elastic constants and yield strength ( $\sigma_y$ ). Yield strength is one of the important properties for device fabrication. ( $\sigma_y$ ) Can be calculated using the relation [37]

$$\sigma_y = H_v/2.9\{[(1-(2-n)][(12.5(2-n)/(1-(2-n)]^{2-n})\}$$

The yield strength is calculated to be 16.72 and 16.941 MPa for  $KSbF_4$  and  $K_2SbF_5$  crystals respectively. The microhardness parameters are listed in Table. 1.

Table 1. Hardness parameters of KSbF<sub>4</sub> and K<sub>2</sub>SbF<sub>5</sub> crystals.

Name of the crystal	n	K <sub>1</sub> (kg/mm)	K <sub>2</sub> (kg/mm)	x(μm)
KSbF <sub>4</sub>	1.59	0.001074	0.02261	0.0777
K <sub>2</sub> SbF <sub>5</sub>	1.63	0.001041	0.02854	0.085

The elastic stiffness constant  $(C_{11})$  gives an idea about tightness of bonding between neighbouring atoms. The stiffness constant for different loads has been calculated using Wooster's empirical formula [38],

$$C_{11} = Hv^{7/4}$$
. (9)

The variation of elastic stiffness constant with load is depicted in Fig. 8. From the graph, it is clear that the stiffness constant increases with increase of load.

The higher value of stiffness constant for K<sub>2</sub> SbF<sub>5</sub> than KSbF<sub>4</sub> may be due to the higher contribution of potassium and fluorine. Appreciable values of C<sub>11</sub> indicates that the binding force between the  ${[Sb_4F_{16}]}^4$  anions and  $K^{\dagger}$  cations in  $KSbF_4$ crystal and [SbF<sub>5</sub>]<sup>2-</sup> anions and K<sup>+</sup> cations in K<sub>2</sub>SbF<sub>5</sub> crystals are quite strong.

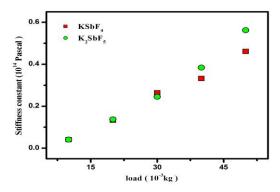


Fig. 8. Variation of stiffness constant with load for KSbF<sub>4</sub> and K2SbF5 crystals

# IV. CONCLUSIONS

Optically clear single crystals of KSbF<sub>4</sub> and K<sub>2</sub>SbF<sub>5</sub> have been grown by slow and controlled evaporation technique. The microhardness study shows that hardness steadily increases with load, then decrease for higher loads. The work hardening coefficient (n) for KSbF<sub>4</sub> and K<sub>2</sub>SbF<sub>5</sub> crystals have been found to be 1.59 and 1.63 respectively and indicate that they are moderately softer substances. Yield strength has been found out as 16.72 and 16.941 MPa for KSbF<sub>4</sub> and K<sub>2</sub>SbF<sub>5</sub> crystals. Stiffness constant revealed that the tightness of binding of ions is quite strong.

# REFERENCES

- [1] M. Senthil Pandian , N. Balamurugan , V. Ganesh, P.V. Rajashekar, K. Kishan Rao and P. Ramasamy, "Growth of TGS single crystal by conventional and SR method and its analysis on the basis of mechanical, thermal, optical and etching studies". Mat. Lett., 62, 2008, pp.3830-3832.
- G. Senthil Murugan, R. Perumal Ramasamy, P. Ramasamy and G. Bh agavannarayana, "Characterization of <0 1 0> directed crystals ammonium malate single grown Sankaranarayanan-Ramasamy method" J. Crys. Growth, 328, 2011, pp 58-61.
- Jianghong Gong,"On the energy balance model for conventional Vickers microhardness testing of brittle ceramics", Mat. Sci. Lett. 19, pp. 515-517.
- R. Robert, C. Justin Raj, S. Krishnan and S. Jerome Das, "Growth, theoretical and optical studies on potassium dihydrogen phosphate (KDP) single crystals by modified Sankaranarayanan-Ramasamy (mSR) method", Physica B 405, 2010, pp. 20-24.
- R. Uthra Kumar, C. Vesta, C. Justin Raj, S. Krishnan, S. Jerome Das, "Bulk crystal growth and characterization of non-linear optical bisthiourea zinc chloride single crystal by unidirectional growth method", Current App. Phys. 10, 2010, pp. 548-552.
- B W Mott, Microindentation Hardness Testing Butterworths, 1956).
- K. J. Prathap and V. Hari Babu, "Microhardness studies in ammonium halide crystals", Bull. Mater. Sci. 2, 1980, pp. 43-53.
- G. Sangaiah and K. Kishan Rao, "Microhardness studies on some alum", crystals Bull. Mater. Sci. 16, 1993, pp. 397-402.
- D. Tabor, The Hardness of Materials (Oxford: Oxford University Press, 1951).
- [10] M.P. Borzenkova, F.V. Kalinchenko, A.V. Novoselova, A.K. Ivanoshits and N.I. Sorokin Rus. J. Inorg. Chem. 29, 1984, pp.703.

- [11] R. L. Davidovich, P.S. Gordienko, J. Grigas, T.A. Kaidalova, V. Urbonavicius and L.A. Zemnukhova, Phys. Stat. Sol. (a) 84, 1984, pp. 387-392.
- [12] A. Cyrac Peter, M. Vimalan, P. Sagayaraj and J. Madhavan Optical, 'Microhardness, Dielectric and Conductivity Studies of Na<sub>2</sub>SbBF<sub>8</sub> Single Organic Crystal", Sch. Res. Lib 2, 2010, pp.191-197.
- Charles,"Microindentation analysis of fluoroantimonate crystals", Mat. Chem. and Phys. 45, 1996, pp.189-192.
- [14] J. Benet Charles and F.D. Gnanam," Vickers micromechanical indentation of NaSb<sub>2</sub> F<sub>7</sub> and Na<sub>3</sub> Sb<sub>4</sub> F<sub>15</sub> single crystals, J. Mater. Sci. Lett. 9, 1990, pp.165-166.
- [15] R. Christudhas, J. Benet Charles and F.D Gnanam J. Mat. Sci. Lett. 12, 1993, pp.1395.
- [16] Rani Christudhas, J. Benet Charles and F.D Gnanam, Growth and microhardness studies on NH<sub>4</sub>Sb<sub>3</sub>F<sub>10</sub> single crystals J. Crys. Growth 137, 1994, pp. 295-298.
- [17] A. Bystroem, S. Baecklund and K.A Wilhelmi, Ark. Kemi. 4, 1952, pp. 175.
- [18] H. H. Udovenko, M.F. Eibelman and R.L. Davidovich Kristallografiya 37, 1992, pp. 735.
- [19] A. Bystroem and K.A Wilhelmi, Ark. Kemi.3,1951,pp. 461.
- [20] Lesley E Alexander and I. R. Beattie, "Vibrational spectra of the isoelectronic species IF<sub>5</sub>, TeF<sup>5-</sup>, and SbF<sub>5</sub><sup>2-"</sup>, J. Chem. Soc. (A) 1971, pp. 3091-3095.
- [21] J. C. Hempel, D. .Klassen, W.E. Hatfield and H.H. Dearman J. Chem. Phys. 58,1973,pp.1487-1494.
- [22] A.G. Kunjomana and K.A. Chandrasekharan, "Microhardness studies of GaTe whiskers", Crys. Res. Tech. 40, 2005, pp. 782-785.
- [23] D. Nagaraju, P.V. Raja Shekar, T. Bhaskar Rao and K. Kishan Rao, Mat. Lett. 64,2010,pp. 267
- V. Ganesh, Snehalatha Reddy Ch, Mohd. Shakir, M. A. Wahab, G. Bhagavannarayana and K. Kishan Rao," Comparative study on BIS thiourea cadmium acetate crystals using HRXRD, etching, microhardness, UV-visible and dielectric characterizations" Physica B, 406, 2011, pp. 259-264.
- [25] K. Sangwal, "On the reverse indentation size effect and microhardness measurement of solids" Mat. Chem. and Phys. 63, 2000, pp. 145-152.
- C.C. Desai and J. L. Rai, Microhardness studies of SnI2 and SnI<sub>4</sub>single crystals" Bull. Mater. Sci.5, 1983, pp. 453-457.
- [27] T.G. Berzina, I.B. Berman and P.A. Savintsev Sov. Phys. Crystallogr. 9,1965, pp. 483
- [28] H. Li, Y.H. Han and R.C. Bradit, "Knoop microhardness of single crystal sulphur", J. Mater. Sci. 29, 1994, pp. 5641 -5645.
- [29] S.K. Arora, G.S. Trivikasama Rao and N.M Batra, "Vickers micromechanical indentation of BaMoO4 crystals" J. Mater. Sci. 19,1984,pp. 297
- [30] H. Li, R.C. Bradit J. Mater. Sci. "The effect of indentation-induced cracking on the apparent microhardness", 31, 1996, pp.1065.
- [31] E. Meyer and Z. Ver Dtsch. Ing., "Analysis of hardness examination and hardness",52,1908,pp. 645-654.
- C. Justin Raj and S. Jerome Das, Crys. Growth & Design, "Bulk Growth and Characterization of Semiorganic Nonlinear Optical l-Alanine Cadmium Chloride Single Crystal by Modified Sankaranarayanan-Ramasamy Method", 8, 2008, pp. 2729.
- [33] S. Jerome Das and R. Gopinathan, Cryst. Res. Tech. 21,1992,pp. K17
- [34] E.M. Onitsch Microscopie, 2,1947,pp. 131.
- [35] Elsamma Chacko, J. Mary Linet, S. Mary Navis Priya, C. Vesta, B. Milton Boaz and S. Jerome Das, "Growth and microhardness studies of mixed crystals of (NH<sub>4</sub>)<sub>2</sub>SbF<sub>5</sub>-K<sub>2</sub>SbF<sub>5</sub>" Ind. J. of Pure and App. Phy. 44, 2006, pp. 260.
- [36] J. Kumar, M. Thirumavalavan, F.D. Gnanam and P. Ramasamy, "Microindentation Studies on Single Crystals of Sodium and Ammonium Fluoroborates" Phys. Stat. Sol. (a),103, 1987,pp. 431.
- R. Ashokkumar, R. Ezhilvizhi, N. Vijayan and D. Rajan Babu Growth, optical and mechanical properties of nonlinear optical alpha-lithium iodate single crystal" Sch. Res. Lib.2, 2010,pp 247.
- [38] W.A. Wooster," Physical properties and atomic arrangements in crystals"

Rep. Progr. Phys. 16, 1953, pp. 62.

Dr. C. Besky Job, Department of Physics and Research Centre, Scott Christian College (Autonomous), Nagercoil, India.

R. Shabu, Department of Physics and Research Centre, Scott Christian College (Autonomous), Nagercoil, India.

Dr. S. Paulraj, Department of Physics and Research Centre, St. Xavier College (Autonomous), Palayamkottai, India